Dedicated to the 80th anniversary of Professor Romanas Višomirskis

Oxygen evolution on composite ruthenium and nickel oxides electrode

Kęstutis Juodkazis*,	Electrochemical behavior of a composite layer of RuO ₂ and NiO deposited on nickel and titani-				
	um electrodes has been studied in acidic, neutral and alkaline media with and without chloride.				
Jurga Juodkazytė,	Cyclic voltammetry was used for the electrochemical studies, whereas the methods of X-ray				
	electron spectroscopy and scanning electron microscopy were employed for the investigation of				
Rasa Vilkauskaitė,	the oxide layer composition and surface morphology.				
<u> </u>	It has been found that the composite layer of RuO2 and NiO deposited thermally on tita-				
Benjaminas Šebeka,	nium electrode has a synergetic catalytic effect on the evolution process of O ₂ in both strongly				
	acidic and strongly alkaline media.				
Vitalija Jasulaitienė	It has been shown that Ti/RuO ₂ ·NiO electrode can be used for obtaining pure oxygen by				
	electrolyzing sea water alkalized to $pH \ge 4$.				
Institute of Chemistry,					
A. Goštauto 9, LT-01108 Vilnius,	Key words: oxygen evolution, ruthenium, nickel, oxide electrode				
Lithuania					

INTRODUCTION

Oxygen evolution reaction (OER) taking place on the surfaces of RuO₂, IrO₂, oxides of iron subfamily metals and other oxides is the object of continuous investigations due to its importance in such areas as seawater electrolysis and energy conversion technologies [1-8]. Researches are being carried out in order to achieve the conditions that would make the process of O₂ evolution technological and as facile as possible. As regards seawater electrolysis, the main challenge is to ensure the purity of oxygen gas preventing the possible contamination with Cl₂. Izumiya et al. have reported [9] that in the case of Ti / IrO₂ / MnO_x electrode in a slightly alkaline medium, O_2 yield can be increased from ~70% to ~90% by introducing Mo oxides into the oxide layer. Among the best known catalysts for OER are ruthenium and iridium dioxides (RuO, and IrO,) and Co and Ni oxides [2]. Mixtures of Ni and Co oxides of NiCo₂O₄ or CoNi₂O₄ type are used in alkaline media [3-6], but they are weaker depolarizers of OER as compared to RuO₂. On the other hand, RuO₂ is more expensive and not sufficiently stable in alkaline and acidic media due to the possible oxidation to soluble Ru(VI), Ru(VII) or volatile Ru(VIII) compounds.

In the present work we have attempted to form a layer of composite RuO_2 and NiO on nickel and titanium substrates with the purpose to evaluate its electrochemical and catalytic proper-

ties for O_2 evolution reaction. Electrocatalytic properties of such mixed oxide in H_2 evolution reaction as well as the possibilities of its application for electrochemical supercapacitors have been investigated in [10–12].

EXPERIMENTAL

Synthesis of mixed oxide layer

The layer of the composite Ru and Ni oxides on titanium substrate was formed as follows: the Ti electrode was etched for 2 min in hot (80 °C) diluted H_2SO_4 (1 : 1), washed with water and blotted with filter paper. Then a layer of a solution containing RuOHCl₃, NiCl₂·6H₂O, HCl, isopropanol and water (pH \approx 0) was painted onto the Ti electrode and dried for several minutes at 65–80 °C. The molar ratio of Ru and Ni in the solution was 1 : 1. Such painting and drying procedure was performed for 5 times and then the electrode was heated additionally for 1 h at 400 °C in air. Approximate thickness of RuO₂ · NiO layer was ~0.5 µm.

The layer of the composite Ru and Ni oxides on nickel substrate was formed as follows: the nickel electrode was degreased with a fine-grained pulp of MgO + Ca(OH)₂, polished with water-proof emery paper, washed and blotted with filter paper. Immediately after this a layer of the solution containing RuOHCl₃, HCl, isopropanol and water (pH \approx 0) was painted onto the nickel surface, dried at ~80 °C and heated for 5 min at 400 °C. Such procedure was performed for 3 times.

Chemicals of analytical grade and triply distilled water were used to prepare the solutions.

^{*} Corresponding author. E-mail: kesjuod@ktl.mii.lt

Electrochemical measurements

The voltammetric measurements were performed using a PI-50–1.1 potentiostat with a PR-8 programming device (Russia) connected to a personal computer. A conventional ~25 cm³ glass cell with separate working, counter and reference electrode compartments was used. A pure Pt (99.99%) plate served as the counter electrode. Hydrogen electrode in a working solution (RHE) was used as the reference electrode. The potential values in the text are referred to RHE scale. The relationship between the two scales is expressed in the following way: $E_{RHE} = E_{SHE} - 0.059$ pH. All the current density values given in the text refer to the apparent area of the working electrode, i. e. 1.0 cm². All the experiments were carried out at room temperature (20 °C).

XPS and SEM measurements

The composition of the composite oxide coatings was analyzed by X-ray photoelectron spectroscopy. The spectra were recorded with a spectrometer "ESCALAB MK II" (VG Scientific, Great Britain) using X-radiation of MgK_{α} (1253.6 eV, pass energy of 20 eV). The spectrometer had a base pressure of $5 \cdot 10^{-8}$ Torr in the analyzer chamber and $2 \cdot 10^{-8}$ Torr in the sample preparation chamber, where Ar⁺ ion bombardment was performed. For sputter etching, the argon gas pressure was maintained at $6 \cdot 10^{-5}$ Torr and the ion gun was operated at 2 kV. The approximate rate of the oxide phase etching was 0.1-0.2 nm min⁻¹ μ A⁻¹.

Quantative elemental analysis was performed by estimating the peak areas and taking into account the empirical sensitivity factors for each element [13], and then the spectra recorded were compared with the standard spectra [14, 15]. The binding energies were calibrated with respect to the C 1s electron peak at 284.6 eV due to the residual pump oil on the sample surface. Not less than 5 spectra were recorded for every element. The maximum accuracy of the method was 0.1%. Standard and experimental binding energy values for nickel and ruthenium compounds are given in Table 1 [14, 15]. A standard program was used for data processing (XPS spectra were treated with a Shirley-type background subtraction and fitted with mixed Gaussian–Lorentzian functions).

Surface morphology of the composite Ru and Ni oxide electrodes was studied using a scanning electron microscope EVO 50 EP (Carl Zeiss SMT AG, Germany) with energy dispersion and wave dispersion X-ray spectrometers (Oxford, UK).

Table. Standard binding energies of nickel and ruthenium compounds



Fig. 1. XPS spectra of mixed $\text{RuO}_2 \cdot \text{NiO}$ layer on Ti substrate: a - as deposited; b - after argon etching



Fig. 2. Scanning electron micrograph of the surface of mixed RuO₂ · NiO layer on Ti substrate

		•					
Compound	Binding energy, eV						
	Ni 2p ^{3/2}		O 1s		Ru 3d		
	Standard	Exp.	Standard	Exp.	Standard	Exp.	
Ni	852.5 ± 0.1	852.6					
Ni(OH) ₂	856.0 ± 0.1	855.8	530.8-531.7	532.0			
NiO	853.3-854.4	854.1	529.3-530.7	529.3 530.7			
Ru					280.0 ± 0.1	279.9	
RuO ₂			529.4	529.3	280.9 ± 0.1	281.0 280.5	
RuO ₃			530.7	530.7	282.5	282.5	



Fig. 3. Cyclic voltammograms of: $1 - \text{Ni} / \text{RuO}_2$ electrode in 0.5 M H₂SO₄: $2 - \text{Ni} / \text{RuO}_2$ electrode in 1 M KOH; 3 - Ni electrode in 1 M KOH; $v = 50 \text{ mV s}^{-1}$, 20 °C



Fig. 5. Cyclic voltammograms of Ti / RuO₂·NiO electrode in: 1 - 1 M KOH, 2-4 - 0.5 M K₂SO₄ + 0.01 M KOH (4 - in stirred solution); v = 50 mV s⁻¹, 20 °C

RESULTS AND DISCUSSION

XPS spectra characterizing the surface state of the composite oxide layer on Ti substrate as-deposited and after argon etching (20 μ A, 30 s), are presented in Figs. 1a and 1b, respectively. The analysis of the spectra as well as standard and experimental binding energy values given in Table 1 shows that the main components of the as-deposited oxide layer are RuO₂, RuO₃, Ni(OH)₂, NiO and NiCl₂ (Fig. 1 a). No RuO₃ was found on the electrode surface after the removal of a thin surface layer (~2 nm) by means of argon etching (Fig. 1 b), whereas some of the metallic Ru and Ni was detected.

Fig. 2 shows a SEM image of $\text{RuO}_2 \cdot \text{NiO}$ coating on the titanium substrate. The micrograph reveals a cracked-mood-looking appearance of the electrode surface, which is typical of thermally formed RuO₂ oxide layers on titanium substrate [7, 16].



Fig. 4. Cyclic voltammograms of Ti / RuO₂ · NiO electrode in: 1 - 0.5 M H₂SO₄, 2 - literature data for Ti / RuO₂ · TiO₂ in 0.5 M Na₂SO₄, pH = 2, 60 °C [7], 3 - 0.5 M K₅SO₄ + 0.01 M H2SO₄, 4 - 0.5 M K₅SO₄ (pH = 5.5); v = 50 mV s⁻, 20 °C

Fig. 3 illustrates the voltammetric behavior of the composite RuO₂ · NiO layer on nickel in the solutions of 0.5 M H₂SO₄ and 1 M KOH. A comparison of curves 1 and 2 with the voltammetric response of pure nickel electrode (curve 3) shows that the addition of RuO₂ to the surface oxide layer on the nickel electrode significantly depolarizes the oxygen evolution process. The positive effect of RuO, is especially pronounced at high anodic current densities. Since nickel is not sufficiently chemically inert, the layer of the composite oxides was formed on titanium substrate, which does not suffer from this disadvantage. A comparison of curve 1 in Fig. 4 with literature data for Ti / $RuO_2 \cdot TiO_2$ electrode [7] (Fig. 4, curve 2) shows that in the case of the strongly acidic solution $(0.5 \text{ M H}_2\text{SO}_4)$, the influence of NiO on the rate of OER is insignificant - up to ~50 mA cm⁻², whereas at higher anodic current densities it becomes evident as the process is depolarized by ~50 mV. The average value of the slope dE / dlgi for curve 1 (Fig. 4) is close to ~60 mV. Similar values of dE / dlgi have been reported in literature [7] for the range of low anodic current densities, i. e. 20-30 mA cm⁻², whereas at higher values of i, dE / dlgi increases considerably. In slightly acidic and neutral media (Fig. 4, curves 3 and 4), the oxygen evolution on the Ti / RuO₂·NiO electrode becomes significantly hindered. The beginning of the process shifts from 1.4 V to ~1.8 V (Fig. 4, curve 3) and even to ~2.0 V (Fig. 4, curve 4) in a slightly acidic and neutral solution of K₂SO₄, respectively. The oxygen evolution process in the case of the positive going part of the cycle is relatively easier, i. e. takes place at lower potentials than in the case of negative going part. This is related to the oxidation of NiO to NiO₂ [17] and the passivation of the electrode surface, as discussed in [18]. NiO₂, presumably, suppresses the electrocatalytic effect of RuO₂, so the use of the Ti / RuO₂ · NiO anode for OER in slightly acidic and neutral media makes no sense. In the case of the strongly acidic medium, NiO₂ is not stable on the electrode surface [17]. As it has been mentioned above, the presence of NiO in the surface oxide layer depolarizes the process of oxygen evolution; however, NiO should not be sufficiently stable under such conditions.

In a strongly alkaline medium, the layer of the composite $\text{RuO}_2 \cdot \text{NiO}$ on Ti substrate is very efficient from the catalytic point of view (Fig. 5, curve 1). OER takes place at 1.5–1.7 V. The average slope dE / dlgi makes ~80 mV up to $i_a \approx 100 \text{ mA cm}^{-2}$. OH⁻ ions should be involved in this process, since OER becomes significantly hindered as the concentration of OH⁻ decreases down to pH 11–12. In slightly alkaline media, the branch of the voltammogram representing OER shifts by ~600 mV towards more positive E values (Fig. 5, curve 2). This should be related to the passivation of the electrode surface like in the case of the acidic medium (Fig. 4), and also to the change in the mechanism of the oxygen evolution reaction.

Curves 3 and 4 in Fig. 5 represent a more detailed view of the processes taking place on the Ti / RuO₂·NiO electrode surface in the range of lower current densities. One can see that the transpassivation process begins at ~1.3 V. According to literature [19], the anodic current wave seen in voltammograms 3 and 4 within *E* range from ~1.3 V to ~1.6 V should be related to the oxidation of NiO and Ni(OH)₂ to NiOOH. The branch of voltammograms 3 and 4 beginning at $E \approx 1.6$ V is related to the catalytic oxidation of O²⁻ ions, which could conceivably be represented by the sequence of the following reactions involving the participation of RuO₂ and NiO recently proposed in [18, 20]:

$$NiOOH + OH^{-} \Leftrightarrow NiO(OH)_{2} + H^{+} + e^{-}$$
(1)

$$\operatorname{NiO(OH)}_{2} + 2OH^{-} \rightarrow \operatorname{NiOO}_{2} + 2H_{2}O + 2e^{-}$$
(2)

$$NiOO_{2} + OH^{-} \rightarrow NiOOH + O_{2} + e^{-}$$
(3)

and

$$\operatorname{RuO}_{2} + \operatorname{OH}^{-} \Leftrightarrow \operatorname{RuO}_{2}^{+} \cdot \left(\operatorname{OH}^{-}\right)_{\operatorname{ad}} + e^{-}$$
(4)

$$\operatorname{RuO}_{2}^{+} \cdot (\operatorname{OH}^{-})_{ad} + \operatorname{OH}^{-} \rightarrow \operatorname{RuO}_{3} + \operatorname{H}_{2}\operatorname{O} + e^{-}$$
(5)

$$\operatorname{RuO}_{3} + 2\operatorname{OH}^{-} \longrightarrow \operatorname{RuO}_{2} + \operatorname{O}_{2} + \operatorname{H}_{2}\operatorname{O} + 2e^{-}$$
(6)

The summary reaction for both cases is:

$$4OH^{-} \rightarrow O_{2} + 2H_{2}O + 4e^{-}$$
(7)

NiOO₂ in eqs. (2) and (3) represents Ni(IV) peroxide, and RuO₃ is Ru(IV) peroxide or Ru(VI) oxide, which form as intermediates in OER [21]. RuO₂⁺ represents a hole in the RuO₂ valence band [20, 22]. When the concentration of OH⁻ ions in the solution is sufficiently high, the oxygen evolution process, according to summary reaction (7), begins at $E \approx 1.5$ V. The above assumption regarding the formation of metal peroxides, which can also be treated as the chemisorption of O₂²⁻ ions on the electrode surface, makes it possible to explain the underpotential occurrence of OER with respect to $E_{H_2O_2/H_2O}^0 = 1.7$ V. In the case of the slightly alkaline solution, the limitation in the rate of the anodic process occurs at $E \ge 1.7$ V, as can be seen from curves 3 and 4 (Fig. 5). Stirring of the solution has some influence on the rate of the process within 1.7–1.9 V (Fig. 5, curve 4). This influence is too weak, however, to presume that the rate of the process is limited by the diffusion of OH^- ions. The limitation of the anodic process rate, like in the case of the neutral medium (Fig. 4, curve 4), should be attributed to the formation of a passivating layer of anhydrous NiO₂ as a result of further oxidation of NiOOH. At E > 2.0 V, the oxygen evolution reaction on the surface of NiO₂ should proceed irreversibly involving the participation of H₂O molecules mainly as suggested in [18]:

$$NiO_{2} + H_{2}O \rightarrow [NiO_{3}] + 2H^{+} + 2e^{-}$$
(8)

$$[\text{NiO}_3] + \text{H}_2\text{O} \rightarrow \text{NiO}_2 + \text{H}_2\text{O}_2$$
(9)

$$H_2O_2 \to O_2 + 2H^+ + 2e^-$$
 (10)

summary equation $2H_2O \rightarrow O_2 + 4H^+ + 4e^-$, (11)

where $[NiO_3]$ is the unstable intermediate in OER. One can see that in the slightly alkaline solution (Fig. 5, curves 2–4), similarly to the slightly acidic and neutral ones (Fig. 4, curves 3, 4), there is no influence of the RuO₂ component on the rate of the process, since OER on the RuO₂ surface in both acidic and alkaline media begins at $E \approx 1.4$ V [21]. As can be judged from the cathodic part of cycles 3 and 4 (Fig. 5), the reduction of NiO₂ to NiOOH and NiOOH to NiO mainly takes place within E range from ~1.8 V to ~1.4 V [19] (Fig. 5, curves 3, 4). The rectangular shape of the voltammograms at E < 1.4 V demonstrates, most likely, the presence of RuO₂ on the electrode surface [21].

It is interesting to note that the addition of 0.5 M NaCl to 1 M KOH has almost no effect on the O_2 evolution process on Ti / Ru O_2 · NiO anode, as can be seen from the comparison of curve 1 in Figs. 5 and 6 a. The slope dE / dlgi makes ~80 mV up to



Fig. 6. Cyclic voltammograms of Ti / $RuO_2 \cdot NiO$ electrode in: 1 - 0.5 M NaCl + 1 M KOH, 2 - 0.5 M NaCl + 0.01 M KOH, 3 - literature data for Ti / $RuO_2 \cdot \text{Ti}O_2$ electrode in ~5 M NaCl, 60 °C [7]; v = 50 mV s⁻¹, 20 °C

 $i_a \approx 100 \text{ mA cm}^{-2}$ in both of the cases. The evolution of Cl₂ under such conditions is possible at $E \approx 2.16$ V in RHE scale ($E_{CLDCL}^0 + 0.059$ pH) [23]. The gap between 1.6 V and 2.16 V is sufficiently large to have a pure O₂ gas evolving in chloride containing electrolyte [24–26]. In the case of the slightly alkaline solution (pH \approx 12), reversible chlorine evolution process is observed at $E \approx 2.04$ V (Fig. 6a, group of curves 2), which is consistent with thermodynamics. The slope dE / dlgi makes up ~60 mV. The analysis of the voltammograms shown in Fig.6 suggests that Cl, evolution process in the latter case (Fig. 6a, curve 2) proceeds reversibly as can be seen from the cathodic part of the cycles. As discussed above (Fig. 5), the process should take place on the surface blocked with NiO, oxide. In order to compare the Cl₂ evolution processes on Ti / RuO₂ · NiO and Ti / RuO₂ · TiO₂ anodes, literature data for Ti / RuO₂ · TiO₂ in ~5 M NaCl, pH= 2.60 °C [7] are presented in Fig. 6 a as curve 3. One can see that the Cl₂ evolution on the surface of RuO₂ · TiO₂ oxide in the slightly acidic medium (pH = 2) takes place at $E \approx 1.5 \text{ V}$ (RHE), which is also consistent with thermodynamics. The slope dE / dlgi makes 35-40 mV, which is close to the value characteristic of the reversible 2 electron transfer process, i. e. 30 mV. The difference between dE / dlgi values reported in literature [7] (Fig. 6 a, curve 3) and obtained by us (Fig. 6 a, curve 2) can be understood presuming that the chlorine evolution takes place on different surface oxides, i. e. mainly on RuO₂ and NiO₂, respectively. One can see that in the range of low current densities, there is no distinct indication of the oxygen evolution process at 1.4-1.5 V in the voltammogram presented in Fig. 6 b. Thus, it can be presumed that in the slightly alkaline medium containing 0.5 M NaCl, mainly Cl- ions adsorb on the electrode surface, preventing the adsorption of OH- ions. When the concentration of hydroxide ions is high enough, only then the adsorption of OH- becomes prevailing and the reactions described by eqs. (1)-(7) can take place.

So, on the basis of the electrochemical study it can be suggested that the titanium electrode with a composite $\text{RuO}_2 \cdot \text{NiO}$ layer can be used to obtain pure oxygen from sea water alkalized to pH \geq 14. Further research is necessary in order to evaluate the technological effectiveness of Ti / RuO₂ \cdot NiO electrode and possible losses of RuO₂ in the course of electrolysis. The use of such electrode for obtaining Cl₂ in slightly alkaline media is inexpedient because of high anodic polarization (>2.0 V) necessary for this process. Besides, it should be noted that chlorine gas may contain some oxygen because OER is also possible under such conditions. Ti / RuO₂ \cdot TiO₂ electrode in the slightly acidic medium can be successfully used for this purpose [7, 27] (Fig. 6, curve 3).

CONCLUSIONS

It has been found that the compound of RuO_2 and NiO oxides thermally deposited on the titanium electrode has a synergetic catalytic effect for the evolution process of O_2 in both strongly acidic and strongly alkaline media.

It has been suggested that Ti / $\text{RuO}_2 \cdot \text{NiO}$ electrode can be used for obtaining pure oxygen by electrolyzing sea water alkalized to pH \ge 14.

> Received 4 December 2007 Accepted 17 December 2007

References

- 1. S. Trasatti, *Electrodes of Conductive Metallic Oxides*, Parts A and B, Elsevier, Amsterdam (1980).
- 2. Y. Matsumoto, E. Sato, Mat. Chem. Phys., 14, 397 (1986).
- Bo Chi, Hong Lin, Jianbao Li, Ning Wang, Jun Yang, Int. J. Hydr. Energ., 31, 1210 (2006).
- Gang Wu, Ning Li, De-Rui Zhou, Kurachi Mitsuo, Bo-Quing Xu, J. Sol. State Chem., 177, 3682 (2004).
- 5. E. B. Castro, S. G. Real, L. F. Pinheiro Dick, *Int. J. Hydr. Energ.*, **29**, 255 (2004).
- Chi-Chang Hu, Yuh-Shu Lee, Ten-Chin Wen, *Mat. Chem. Phys.*, 48, 246 (1997).
- 7. J. Aromaa, O. Forsen, *Electrochim. Acta*, **51**, 6104 (2006).
- N. V. Korovin, E. V. Kasatkin, *Elektrokhimiya*, 29, 448 (1993).
- K. Izumiya, E. Akiyama, H. Habazaki, N, Kumagai, A. Kawashima, K. Hashimoto, *Materials Transactions, JIM*, 38, 899 (1997).
- L. Vazquez-Gomez, S. Cattarin, P. Guerriero, M. Musiani, Electrochim. Acta, 52, 8055 (2007).
- F. Pico, J. Ibanez, T. A. Centeno, C. Pecharroman, R. M. Rojas, J. M. Amarilla, J. M. Rojo, *Electrochim. Acta*, **51**, 4693 (2006).
- 12. X. M. Liu, X. G. Zhang, Electrochim. Acta, 49, 229 (2004).
- D. Brigs, M. P. Seach (eds.), Practical surface analysis by Auger, X-ray photoelectron spectroscopy, Mir, Moscow (1987).
- C. D. Wagner, W. M. Riggs, L. E. Davis, J. F. Moulder (et al.), Handbook of X-ray Photoelectron Spectroscopy, Minnesota 55344, Perkin-Elmer (1978).
- C. D. Wagner, A. V. Naumkin, A. Kraut-Vass, J. W. Allison, C. J. Powell, J. R. Rumble Jr., *NIST Standard Reference Database 20, Version 3.4 (Web Version).*
- 16. A. J. Terezo, E. C. Pereira, Mat. Lett., 53, 339 (2002).
- 17. M. Pourbaix, *Atlas d'équilibres électrochimiques*, Paris, Gauthier-Villars (1963).
- K. Juodkazis, J. Juodkazytė, R. Vilkauskaitė, V. Jasulaitienė, J. Sol. St. Electrochem., DOI: 10.1007/s10008-007-0484-0.
- M. Wherens-Dijksma, P. H. L. Notten, *Electrochim. Acta*, 51, 3609 (2006).
- K. Juodkazis, J. Juodkazytė, V. Šukienė, A. Grigucevičienė, A. Selskis, J. Sol. State Electrochem., DOI: 10.1007/ s10008-007-0476-0.
- J. Juodkazytė, R. Vilkauskaitė, B. Šebeka, K. Juodkazis, Trans. IMF, 85, 194 (2007).
- P. S. Patil, A. Ennaoui, C. D. Lokhande, M. Muller, M. Giersig, K. Diesner, H. Tributsch, *Thin Solid Films*, **310**, 57 (1997).
- 23. Y. Y. Lurje, *Handbook of Analytical Chemistry*, Khymia, Moscow (1979).
- 24. P. C. S. Hayfield, Plat. Met. Rev., 42, 27 (1998).
- 25. P. C. S. Hayfield, Plat. Met. Rev., 42, 46 (1998).
- 26. P. C. S. Hayfield, Plat. Met. Rev., 42, 116 (1998).
- V. V. Gorodeckiy, M. M. Pecherskiy, V. B. Yanke, N. Ya. Bune, V. B Busse-Machukas, V. L. Kubasov, V. V. Losev, *Elektrokhimiya*, 17, 513 (1981).

Kęstutis Juodkazis, Jurga Juodkazytė, Rasa Vilkauskaitė, Benjaminas Šebeka, Vitalija Jasulaitienė

DEGUONIES SKYRIMASIS ANT MIŠRAUS RUTENIO IR NIKELIO OKSIDŲ ELEKTRODO

Santrauka

Darbe tirta mišraus RuO₂ ir NiO oksidų sluoksnio, suformuoto ant nikelio arba titano paviršiaus, elektrocheminė elgsena rūgščiuose, neutraliuose bei šarminiuose tirpaluose su chlorido jonais arba be jų. Elektrocheminiams tyrimams naudota ciklinė voltamperometrija. Oksidinio sluoksnio sudėtis bei jo paviršiaus morfologija tirta rentgeno fotoelektroninės spektroskopijos bei skleidžiamosios elektroninės mikroskopijos metodais. Nustatyta, kad mišrus rutenio ir nikelio oksidų sluoksnis, termiškai suformuotas ant titano pagrindo, depoliarizuoja deguonies skyrimosi procesą tiek labai rūgščioje, tiek labai šarminėje terpėje. Parodyta, kad Ti / RuO₂ · NiO elektrodas gali būti panaudotas grynam deguoniui gauti iš jūros vandens, pašarminto iki pH \geq 14.